Photoluminescence studies in swift heavy ion bombarded mullite

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Abstract

Photoluminescence (PL) of pelletized mullite samples irradiated with 100 MeV swift Ni\textsuperscript{8+} ions with fluences in the range $1 \times 10^{11} - 5 \times 10^{13}$ ions/cm\textsuperscript{2} have been studied at RT. A broad and strong PL emission band with peak at $\sim 537$ nm besides a sharp emission band with peak at $\sim 705$ nm have been recorded in combustion synthesized mullite. However, a single broad emission with peak at $\sim 537$ nm have been recorded in mullite synthesized by sol–gel technique at an excitation of 442 nm laser beam. It is found that PL intensity is decreases with increase of Ni\textsuperscript{8+} ion fluence. The decrease in PL intensity is attributed to destruction of Al–O and Si–O bonds in the samples. Infrared absorption (IR) studies of as it is and Ni\textsuperscript{8+} ion irradiated mullites have also been studied and results obtained are discussed.

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1. Introduction

Good amount of work, though not exhaustive, with heavy ion beams on alkali halides and other materials has been reported [1–5]. The energy loss mechanisms for low energy heavy ions have been well studied [6]. High-energy ions upon entering a medium loses its energy initially through the process of electronic excitation and ionization known as electronic energy loss, and at the end of its range, loses its energy mostly through the process of nuclear loss, in which the lattice atoms of the medium are displaced due to nuclear elastic collisions. The later phenomenon of energy loss becomes prominent when the ion energy reduces to $\sim X$ keV, where $X$ is the atomic weight of the ion. In the surface region of the sample through which an ion enters, the energy loss suffered by a heavy ion through electronic excitation is greater by three orders of magnitude than that through nuclear elastic collisions. In this way the energy deposited in the surface region through elastic collisions should have negligible contribution in defect formation. As such in the sample exposed to high-energy heavy ions, the atoms in the surface

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region are not expected to get displaced from their lattice positions. The other processes through which defects may be induced to heavy ions are due to Coulomb explosion [7] and thermal spike [8].

In this paper we report the PL and IR results of mullite irradiated with 100 MeV Ni$^{8+}$ ion for fluences in the range $1 \times 10^{11} - 5 \times 10^{13}$ ions/cm$^2$. Mullite is an aluminum silicate ($3\text{Al}_2\text{O}_3 \cdot 2\text{SiO}_2$) crystallizes in orthorhombic. This is a promising material for high temperature applications because of its low thermal conductivity. The silicate group of this mineral used in the production of non-fusion cast tank blocks, in the manufacture of spark plugs and as a mineral specimens [9].

2. Experimental

Nano-size particles of mullite are synthesized by combustion (sample I) and sol–gel (sample II) methods. The details of synthesis of both the methods are given elsewhere [10,11]. Pellets of 1 mm thickness and 6 mm diameter are prepared using agate and mortar and polyvinyl alcohol as binder and by applying a pressure of 6 tones per ram of 6 mm diameter using an home made pelletizer. The pellets of sample I and II are bombarded with 100 MeV Ni$^{8+}$ ions from 16 MV pelletron at Nuclear Science Center, New Delhi. For in situ PL measurements a Kimmon He–Cd laser through a sapphire window is used. The photons of sample I and II are bombarded with 100 MeV Ni$^{8+}$ ions for fluences in the range $1 \times 10^{11} - 1 \times 10^{14}$ ions/cm$^2$ followed by 442 nm laser beam excitation are recorded and the results obtained are presented in Figs. 2 and 3, respectively. A broad emission band with peak at ~557 nm besides a sharp emission band with peak at 705 nm is observed in sample I. However, a single emission band with peak at about 537 nm is recorded in sample II. The PL intensity in both samples I and II are found to decrease with increase of ion fluence as shown in Fig. 4. This may be attributed to irradiation induced amorphization as a result of cascade quenching [5,12]. Amorphization may take place in the system as each incident ion may create one or several displacement cascades, which become amorphous as a result of rapid quenching, and these cascades eventually overlap to form an amorphous solid [5]. The susceptibility of amorphization may be measured in two ways: (i) the ion dose required for

3. Results and discussion

Photoluminescence of mullite samples I and II irradiated with 100 MeV Ni$^{8+}$ ions for fluences in the range $1 \times 10^{11} - 1 \times 10^{14}$ ions/cm$^2$ followed by 442 nm laser beam excitation are recorded and the results obtained are presented in Figs. 2 and 3, respectively. A broad emission band with peak at ~557 nm besides a sharp emission band with peak at 705 nm is observed in sample I. However, a single emission band with peak at about 537 nm is recorded in sample II. The PL intensity in both samples I and II are found to decrease with increase of ion fluence as shown in Fig. 4. This may be attributed to irradiation induced amorphization as a result of cascade quenching [5,12]. Amorphization may take place in the system as each incident ion may create one or several displacement cascades, which become amorphous as a result of rapid quenching, and these cascades eventually overlap to form an amorphous solid [5]. The susceptibility of amorphization may be measured in two ways: (i) the ion dose required for
amorphization at a fixed irradiation temperature or (ii) the critical temperature. Wang et al. [5] have observed the complete amorphization in Al₂SiO₅ by monitoring the selected area diffraction technique (SAD) under 1.5 MeV Xe⁺ ions with the fluence of 1.88 × 10¹⁴ ions/cm², and the critical temperature for amorphization of kyanite was observed to be 1281 K. It is also found that the PL intensity in sample I is more compared to that in sample II. This is attributed to the physical nature of the sample, preparation conditions and/or ingredients used in the preparation of phosphor.

The infrared absorption of as it is and that of ion irradiated mullite (I and II) with a fluence of 1 × 10¹³ ions/cm² have been studied and the results obtained for combustion synthesized mullite are given in the Fig. 5 as illustration. The data obtained from IR results on these samples (I and II) are given in the Table 1. The spectra reveals the characteristic absorption bands of silicates and water before irradiation of the samples. It is observed that the sharpness of the peaks is high in ion irradiated samples. This could be attributed to close packing of the molecules or to conformational changes in the molecules or both [13]. The
decrease in the PL intensity in both the samples might be due to the destruction of Si–O and Al–O bonds. And, the destruction of these bonds with irradiation may further enhance the amorphous nature of the sample. The irradiation effects may lead to the restructuring of the surface chemical species because of the energy deposited through electronic loss during the process of swift heavy ion irradiation and, formation of ion induced defects leading to non-radiative recombination centers. These two processes are simultaneous consequences of irradiation and they compete with each other. Consequently, the enhancement or degradation of PL might be due to the balance between these two effects [14].

4. Conclusion

The decrease in PL intensity with increase of ion fluence might be due to destruction of Al–O and Si–O bonds in the present samples. However, the enhancement in the PL intensity in combustion synthesized mullite (I) compared to that in sol–gel technique (II) is attributed to the process of preparation of the sample, which leads to variation in particle size.

Acknowledgements

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References


Table 1

<table>
<thead>
<tr>
<th>Sample</th>
<th>Absorption peak (cm⁻¹)</th>
<th>Unirradiated sample</th>
<th>Irradiated sample</th>
</tr>
</thead>
<tbody>
<tr>
<td>Mullite (I)</td>
<td>467–472 Si–O (γ₄)</td>
<td>Si–O (γ₄)</td>
<td></td>
</tr>
<tr>
<td></td>
<td>566–571 Si–O–Al</td>
<td>Si–O–Al</td>
<td></td>
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<tr>
<td></td>
<td>747–829 Al–O</td>
<td>Al–O</td>
<td></td>
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<tr>
<td></td>
<td>1109–1114 Si–O (γ₃)</td>
<td>Si–O (γ₃)</td>
<td></td>
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<tr>
<td></td>
<td>1160 Si–O (γ₂)</td>
<td>–</td>
<td></td>
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<tr>
<td></td>
<td>2376–2929 Si–O (2γ₅)</td>
<td>Si–O (2γ₅)</td>
<td></td>
</tr>
<tr>
<td></td>
<td>3441–3461 –OH (γ₁, γ₃)</td>
<td>–OH (γ₁, γ₃)</td>
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<tr>
<td>Mullite (II)</td>
<td>447–467 Si–O (γ₄)</td>
<td>Si–O (γ₄)</td>
<td></td>
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<tr>
<td></td>
<td>591–638 Al–O</td>
<td>–</td>
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<tr>
<td></td>
<td>1093–1103 Si–O (γ₃)</td>
<td>Si–O (γ₃)</td>
<td></td>
</tr>
<tr>
<td></td>
<td>2350–2850 Si–O (2γ₅)</td>
<td>–</td>
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</tr>
<tr>
<td></td>
<td>3452–3462 –OH (γ₁, γ₃)</td>
<td>–OH (γ₁, γ₃)</td>
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</table>